The determination of when a body will fail by the growth of a dominant crack requires two things

- . The determination of the stress and strain fields in the body or at least. some mechanical quantity (eg., energy) that can characterize the intensity of loading This requires solving <sup>a</sup> boundary value problem
- . A criterion for the crack to advance/propagate. (This distinguishes this course from what you have learnt from linear/nonlinear elasticity).

We will discuss the bup and the criterion extensively in the class. However, let's first look at an ideal, simplified problem - the strength measured by stretching an infinite atomic lattice to two separated semi-infinite latices:

I gang you 0 <sup>0</sup> <sup>0</sup> 0 0 0 <sup>0</sup> 0 00 fao <sup>9</sup> <sup>8</sup> <sup>0</sup> <sup>0</sup> <sup>0</sup> <sup>a</sup> 5 Got83 <sup>0</sup> <sup>0</sup> <sup>0</sup> <sup>0</sup> <sup>0</sup> <sup>0</sup> <sup>0</sup> <sup>0</sup> <sup>0</sup> 0 dots <sup>0</sup> <sup>0</sup> <sup>0</sup> <sup>0</sup> <sup>0</sup> <sup>0</sup> <sup>0</sup> 0 0 0 0 t 6 <sup>6</sup>

The  $6 - 8$  curve looks like:



The stress state are identical between  $\oplus$  and  $\oplus$ . But there is a difference in energy

$$
2\gamma = \int_{0}^{\infty} x \, d\theta
$$
  
\n $\int$  surface energy required to create 2 new surfaces

Let's estimate Speak wring a "back of the envelop" type of approximation and some material properties that we are familar with. Assuming the following form of  $4 - 8$  relation:

$$
\Delta = A \, \delta \, e^{-\Delta/\beta}
$$

which appears  $6 - 8$  as  $8 \ll 8$  and  $5 - 8$  as  $8 \gg 8$ . To determine A,  $\beta$ , we can use

$$
\frac{d\delta}{d\delta}\Big|_{\delta=0} = \frac{E}{a_0} \Rightarrow A = \frac{E}{\delta}.
$$
\n
$$
\int_0^\infty A \delta e^{-\delta/\beta} = 2\delta \Rightarrow A \Big( -\beta \delta e^{-\delta/\beta} - \beta^2 e^{-\delta/\beta} \Big) \Big|_0^\infty = 2\delta \Rightarrow \beta = \left( \frac{2\delta a_0}{E} \right)^{1/2}
$$

$$
\Rightarrow \frac{d\delta}{d\delta}\Big|_{\delta} = 0 \Rightarrow \delta_{peak} = \beta
$$
  
 
$$
\delta_{peak} = \delta(\delta = \delta_{peak}) = \frac{\sqrt{2}}{e} \left(\frac{\partial E}{\partial \rho}\right)^{1/2} \simeq \frac{1}{2} \left(\frac{\partial E}{\partial \rho}\right)^{1/2}
$$

Some typical values:

$$
E \sim 100 GPa, \quad \Omega_0 \sim 1 \text{ Å}, \quad \Upsilon \sim 1 \text{ J/m}^2
$$
  

$$
\Rightarrow \text{Spec} k \sim \frac{1}{2} \left(\frac{1 \times 10^{11}}{10^{-10}}\right)^{1/2} Pa \approx 15 GPa
$$

This is <sup>a</sup> very high strength Only for nearly perfect crystals i.e graphen Speak Elio is the correct order of magnitude More generally it is to <sup>E</sup> The question now is why materials are so weak on why Elio has been an overestimation Flaws

 $\bigcirc$ 

strength of materials accounting for flaus

Recall from elasticity theory that the stress concentration near an eliptical I Gapp 个 flow is given as Gmax  $\int$  2 b  $\frac{S_{\text{max}}}{S_{\text{app}}}$  =  $1+2\frac{C}{b}$  $\overline{e}$  $\overline{2C}$  $-$ Inglis (1913)  $\int \int \int \int \int \int$  d d sapp

The radius of curvature of the ellipse is given as 
$$
e = \frac{b^2}{c}
$$
. We then have  $\int_{\frac{1}{c} \sqrt{c}}^{c} k(s) = \frac{|a'(s) \times a''(s)|}{|a'(s)|^3}$ 

Now change the failure criterion from Sapp=Speak to Smax=Speak, i.e.,  $26app \left(\frac{c}{\ell}\right)^{l_2} = \frac{1}{2}\left(\frac{E \tau}{a_0}\right)^{l_2} \rightarrow \left\{ \frac{1}{2} \left( \frac{E \tau}{a_0} \right)^{l_2} \right\} = \frac{1}{4}\left(\frac{E \tau}{c} \cdot \frac{\ell}{a_0}\right)^{l_2} \approx \frac{1}{4}\left(\frac{E \tau}{c}\right)^{l_2}$ for the sharpest crack/flaw that is likely to propagate first.

- If we take  $c \sim 1 \mu m$ , we can have  $\sigma_{\text{failure}} \sim 75 \text{ M/s} \cdot \frac{E}{1000}$ . This is close to the range for materials we are familar with.
- . We have assumed the material does not deform plastically. Hence this model is most applicable to rocks, glasses and ceramics at relatively low temperature
- . Note the  $1/5c$  dependence of the strength. For brittle materials, failure strength is <u>not</u> a material property. How to measure? - Introduce notches that are large enough

Griffith theory An energy approach

Assumptions: Linear elastic, isotropic, homogeneous, perfectly brittle, small cracks, plane strain

 $\bigcirc$ 



We want to find the potential energy of the system with respect to crack length  $l$ . Consider the problem as the superposition of problem  $0$ and problem 2

The total potential energy of the system is given as w Incorrect  $v^0 + W_{SE} - W_{S} = W - \frac{1}{2}W_{S}$ Vincanty in general

The total energy of the system oper thickness is  $\phi = W + 2\gamma l = W^{\circ} - \frac{1}{2}W_{\gamma} + 2\gamma l$ since it is energy of a uniform stress problem independent of  $l$ 

To determine  $W_S$ , recall from elasticity theory that the crack opening displ. is given as

$$
\oint = \frac{4\angle(\vdash \nu^2)}{E} \sqrt{\frac{\ell^2}{4} - x^2}
$$

We will also derive this by solving the BVP shortly. The north clone by  $\varsigma$  in the auxiliary problem 2 is then simply

$$
W_{s} = \int_{-\frac{1}{2}}^{\frac{1}{2}} 4\delta \, dx = \frac{4\xi^{2}(1-\nu^{2})}{E} \int_{-\frac{1}{2}}^{\frac{1}{2}} \sqrt{\frac{l^{2}}{4} - x^{2}} dx = \frac{\pi}{2} \frac{1-\nu^{2}}{E} \xi^{2} \ell^{2}
$$

Chatgdp gives covered steps but a wrong answer by a factor of 2 ...

\n
$$
\int_{-\frac{1}{2}}^{\frac{1}{2}} \sqrt{\frac{R^2}{4} - x^2} dx \xrightarrow{\frac{X = \frac{1}{2} \sin \theta}{\frac{1}{2} \cos \theta} \frac{\pi}{\theta}} \int_{-\frac{\pi}{2}}^{\frac{\pi}{2}} \frac{\frac{1}{2} \cos^2 \theta}{4} d\theta = \frac{1}{8} l^2
$$

Back to the total energy of the system



. The system will be in thermodynamic equilibrium (not stable though) when

$$
\frac{\partial \phi}{\partial \ell}\Big|_{\zeta} = 0 \to 2\Upsilon - \frac{\pi}{2} \frac{L^2}{E} \zeta^2 \ell_c = 0 \to \ell_c = \frac{4}{\pi} \frac{\Upsilon E}{(1-\nu^2)\zeta^2}
$$

- . Under prescribed  $6$ , cracks with  $25\ell$  (i.e.,  $\frac{3\phi}{2l}$  $\ll$ o) will grow while cracks with  $l < l_c$  (i.e.,  $\frac{3\phi}{\delta l} > o$ ) will heal". In air, this doesnot actually happen because once a crack forms, a barrier such as oxide and passivation layer is created spontaneously In vacuum crack healing can and does occur. For instance, in space, designers have to deal with the problem of cold welding
- . For a given  $l$ , the critical stress for the crack to propagate is

$$
\mathcal{S}_{\mathsf{c}} = \left( \frac{4}{\pi} \frac{\mathsf{F}}{1 - \nu^2} \cdot \frac{\mathcal{F}}{\ell} \right)^{1/2},
$$

which again leads to a VIT dependence of strength on crack length (Nite that this is based on an energy approach).

Define the "driving force" for crack propagation—energy release rate 
$$
\frac{G}{G} = -\frac{\partial W}{\partial A} = -\frac{\partial Wz}{\partial (lvt)}
$$
 Potential energy  
\nSuch that crack propagation occurs when  $G = 2\delta$  for perfectly birthel's solids.

"Resisting force"

 $\emptyset$ 

The crack growth is unstable when  $\partial \mathcal{G}/\partial l$  >0  $\int \partial \mathcal{G}/\partial l = \pi \sigma^2 (H^{\nu})/E$  for this problem. Alternatively, the crack growth is stable when  $\partial\mathcal{G}/\mathcal{H}<\mathcal{D}$  and neutrally stable when  $\partial\theta/\partial l = o$ .

Let's return to our calculation of the strain energy in our problem Under given identical BCs, we have  $6x - 6y + 6$ , and  $W \neq W^0 + W^0$ . Why did it work for our problem?



$$
W_{SE} = \int_{V} \frac{1}{2} \delta_{ij} \epsilon_{ij} dV = \int_{S} \frac{1}{2} T_{i} u_{i} ds \quad (by \text{ principle of virtual work})
$$
  
= 
$$
\int_{S} \frac{1}{2} (T_{i}^{0} + T_{i}^{0}) (u_{i}^{0} + u_{i}^{0}) ds
$$
  
= 
$$
W_{SE}^{0} + W_{SE}^{0} + \int_{S} \frac{1}{2} T_{i}^{0} u_{i}^{0} ds + \int_{S} \frac{1}{2} T_{i}^{0} u_{i}^{0} ds
$$
  
The two are identical by recognized theorem

Let's evaluate  $\int_{S} \frac{1}{2} T_{i}^{\circledcirc} u_{i}^{\circledcirc} ds$  since  $T_{i}^{\circledcirc}$  is simply  $\circledcirc$  at  $S = S_{\infty}$ .  $\int_{S} \int_{\cdot}^{\cdot} u_{i}^{0} ds = \int_{S^{+}} T_{i}^{0} u_{i}^{0} ds + \int_{S^{-}} T_{i}^{0} u_{i}^{0} ds = 0$ since  $T_i^{\bigcirc}$  on  $S_t = T_i^{\bigcirc}$  on  $S$ - while  $u_i^{\bigcirc}$  on  $S_t = u_i^{\bigcirc}$  on  $S$ -.

 $\Rightarrow$   $W_{SE}$  =  $W_{SE}^{\Phi}$  +  $W_{SE}^{\Phi}$  for our problem. This is not true in general. However, we have had this for  $W_{SE} = W_{Bending} + W_{Stretoting} + W_{Torsion}$ .

Energy release rate

Recall : 
$$
G = -\frac{\partial W}{\partial l}
$$
 (in 20)  
\n $W =$  stored strain energy - work done by loads (per unit  
\n $l =$  total crack length  
\n $l =$  total crack length

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Consider a non-linear elastic body uith a crack subject to point loading.



Sine  $U = \int_{0}^{8} P d\zeta$ ,  $P = \frac{\partial U}{\partial \zeta}$ . Besides,  $U(\zeta, l)$  is a function of both  $\zeta \&l$ . Now consider a specimen with a crack length of  $l+al$ . The structural behavior will be "softer".



$$
W(L) = OBE, \quad W(l+\Delta l) = ODE, \quad \zeta = -\Delta W/\Delta l = OBD/\Delta l.
$$
\n
$$
W(L) = U(\zeta, L)
$$
\n
$$
W(L) = U(\zeta, L)
$$
\n
$$
W(l+\Delta l) = U(\zeta, l+\Delta l) = U(\zeta, l) + \frac{\partial U}{\partial l} \Big|_{\zeta} \Delta l + O(\Delta l^{2})
$$
\n
$$
\Delta W = W(l+\Delta l) - W(l) = \frac{\partial U}{\partial l} \Big|_{\zeta} \Delta l + O(\Delta l^{2})
$$
\n
$$
S = \lim_{\Delta l \to 0} -\frac{\Delta W}{\Delta l} = -\frac{\partial U}{\partial l} \Big|_{\zeta}
$$

· Load control

 $W(k) = -0AB$ ,  $W(l+a\ell) = -0AC$ ,  $G = -\Delta W/a\ell = 0Bc/a\ell$  [will show  $Bcp\sim \Omega^2$ ].  $W(l) = U(\xi, l) - \beta \xi$  $W(l+2l) = U(\delta+2\delta, l+2l) - P(\delta+2\delta)$ =  $U(\frac{1}{2}) + \frac{2U}{\frac{351}{2}} = \frac{2U}{3} = 0.5 - 0.61^2$  $N$ ote  $4\delta = \frac{3\delta}{2l}$  al  $\Delta W = \frac{\partial U}{\partial l}\Big|_{S} \Delta l + \Delta (\Delta l^{2})$  $\Rightarrow$   $G = \lim_{\Delta l \to 0} -\frac{\Delta W}{\Delta l} = -\frac{\partial U}{\partial l}\Big|_{\delta}$  (so BCD- $\Delta l^2$ , back to the  $\{i\}$  mume)

For point loading, G is the same under load or displacement control. For more general loading it can also be shown that G does not depend on the loading conditions. On the other hand, crack growth statolity does depend on them!



We can treat the specimen as two contilever beams of length l.  $\bigcup_{n=1}^{\infty} P\delta = \bigcap_{n=1}^{\infty} \frac{1}{2} \delta \xi dV$  (Note  $\zeta = \frac{Mq}{\pm}$ ,  $\zeta = \frac{\delta}{\pm}$ )  $=\int_{0}^{x} \frac{1}{2} \frac{M^{2}}{rT^{2}} y^{2} dV$  $=\int_{0}^{R} \frac{1}{\gamma} \frac{M^{2}}{FT} dx$  (or  $\int_{0}^{R} \frac{1}{2}MKdx$ ,  $M=Px$ )  $=\frac{1}{6}\frac{\rho^2 l^3}{\Gamma T}=\frac{2 \rho l^3}{\Gamma L L^3}$  $\Rightarrow$   $p = \frac{1}{4} \frac{Ebh^3}{B^3} \cdot \delta$ Fixed P  $W = 2U - 2P_0 = -\frac{4P^2R^3}{E+1.8}$  $G = \frac{3W}{2(hl)} = \frac{12p^2l^2}{\pi l^2l^3}$  $\circ \rightarrow$  Unstable

Fixed 8 South of State 8 is fixed Or we can set the reference energy at  $W = 2U - W$ ork done by this level of 8  $\Longleftrightarrow$  $=\frac{4P^{2}\ell^{3}}{\Gamma+h^{3}}\left(-\frac{4\ell^{5}}{\Gamma b h^{3}}\frac{1}{16}\frac{(\Gamma b h^{3})^{2}}{b^{6}}\delta^{2}\right)$  $=$   $\frac{1}{4}$   $\frac{Ebh^{3}}{03}$   $\zeta^{2}$  $G = -\frac{\partial W}{\partial (l b)}\Big|_{c} = \frac{3}{4} \frac{Eh^{3}}{l^{4}} \delta^{2} = \frac{3}{4} \frac{Eh^{3}}{l^{4}} \Big| 16 \frac{l^{6}}{(Eh^{3})^{2}} p^{2} \Big| = \frac{12 p^{6} l^{2}}{Eh^{3}} \sqrt{16 \frac{2}{h^{6}} l^{2}}$ 

stersting. St 16

Stability : 
$$
\frac{\partial G}{\partial l}\Big|_{\delta} = -3 \frac{Eh^3 \delta^2}{l^5} < 0 \rightarrow
$$
 Stable !

Let's move on to experimentally accessible quatitaties, including P.S.L.



- $L_1$  Introduction
- L2 + 23 & HW1

To show the concept of Griffith's theory further let's consider the following form of analysis.  $x + x$ 

$$
\frac{f^{p}d}{f} = \int_{0}^{R} \frac{1}{2}EI(k^{2}d) + T^{p} - Pd
$$
\n
$$
\frac{f^{p}d}{f} = \int_{0}^{R} \frac{1}{2}EI(w^{n})^{2}d + T^{p}
$$
\n
$$
\frac{1}{2} \int_{0}^{R} \frac{1}{2}EI(w^{n})^{2}d + T^{p}
$$
\n
$$
\frac{1}{2} \int_{0}^{R} \frac{1}{2}EI(w^{n})^{2}d + T^{p}
$$
\n
$$
\frac{1}{2} \int_{0}^{R} EI(w^{n})d + T^{p}
$$
\n
$$
\frac{1}{2}EI(w^{n})\{1 + \frac{1}{2}EI(w^{n})\} + T^{p}
$$
\n
$$
\frac{1}{2}Li(w^{n})\{1 + \frac{1}{2
$$

$$
\Rightarrow \delta F = \int_{0}^{2} ET \, \omega^{\text{un}} \delta \omega \, dx + EL \omega^{\text{un}} \, \delta \omega^{\text{un}} \big|_{R} - EL \, \omega^{\text{un}} \, \delta \omega^{\text{un}} \big|_{0} - EL \omega^{\text{un}} \, \delta \omega \big|_{R} + EL \omega^{\text{un}} \, \delta \omega \big|_{0} + EL \omega^{\text{un}} \, \delta \omega \big|_{0}
$$
\n
$$
+ \left( \frac{1}{2} ET \, \omega^{\text{un}} \, (l) + T \right) \, \delta l.
$$

Note 
$$
\int w(k) = \int w|_{\ell} + w \cdot \int k
$$
.  $\rightarrow \int w(k) = \int w'|_{\ell} + w''(k) \cdot \int k = 0$  (since w(k)=0)

$$
Sw|_{R}
$$
\n
$$
Sw|_{R}
$$
\n
$$
Sw|_{L}
$$
\n
$$
Sw = \int_{0}^{R} E L w''' \, Sw dx + (\Gamma - \frac{1}{2} E L w''(l)) \, ds
$$
\n
$$
Sw = \int_{0}^{R} E L w''' \, Sw dx + (\Gamma - \frac{1}{2} E L w''(l)) \, ds
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$$
Sw = \int_{0}^{R} E L w''' \, Sw dx + (\Gamma - \frac{1}{2} E L w''(l)) \, ds
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$$
\n
$$
Sw = \int_{0}^{R} E L w''' \, Sw dx + (\Gamma - \frac{1}{2} E L w''(l)) \, ds
$$

Mathematica: << Variational Methods

Euler Equations 
$$
\left[ \frac{1}{2}EL(w^{\prime\prime}Lx)\right]^{2} - \rho \cdot WLx
$$
,  $WLxL \times$ ]  
These not change the local criterion

Questions: 1 What if there is externed loading PCx)

2 What if there are both bending and forston

This Griffith concept is not limited. You would find a wide range of problems that accounting for surface energies.



(AI)

$$
(3) = \frac{6\frac{w^{2}}{\sqrt{1 + w^{2}}} \times 6w \Big|_{\alpha} - \frac{w^{2}}{\sqrt{1 + w^{2}}} \times 6w \Big|_{\alpha}
$$
\n
$$
= -\frac{8w \Big|_{\alpha} + w^{2}(\alpha) \times 6w^{2}}{\sqrt{1 + w^{2}(\alpha)}} \text{ as } 6\alpha
$$
\n
$$
= -\frac{8w \Big|_{\alpha} + w^{2}(\alpha) \times 6w^{2}}{\sqrt{1 + w^{2}(\alpha)}} \text{ as } 6\alpha
$$
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$$
= -\frac{8w \Big|_{\alpha} + w^{2}(\alpha) \times 6w^{2}}{\sqrt{1 + w^{2}(\alpha)}} \text{ as } 6\alpha
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= -\frac{8w \Big|_{\alpha} + w^{2}(\alpha) \times 6w^{2}}{\sqrt{1 + w^{2}(\alpha)}} \text{ as } 6\alpha
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\n
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= -\frac{8w \Big|_{\alpha} + w^{2}(\alpha) \times 6w^{2}}{\sqrt{1 + w^{2}(\alpha)}} \text{ as } 6\alpha
$$
\n
$$
= -\frac{8w \Big|_{\alpha} + w^{2}(\alpha) \times 6w^{2}}{\sqrt{1 + w^{2}(\alpha)}} \text{ as } 6\alpha
$$
\n
$$
= -\frac{8w \Big|_{\alpha} + 6w \Big|_{
$$

Arbitrary 
$$
\delta\omega
$$
,  $\delta a \Rightarrow \Delta \beta = \sqrt{\frac{w^4}{(1+w^2)^{3/2}} + \frac{w^1}{r(nw^2)^{3/2}}}$ ,  $\cos \theta = \frac{\gamma_{sr} - \gamma_{s1}}{\gamma}$ 

\nNow  $\gamma = \sqrt{\frac{1}{\gamma_{s1} - \gamma_{s2}}}$ ,  $\frac{\gamma_{s2}}{\gamma_{s2} - \gamma_{s1}}$ 

\nNow  $\gamma = \sqrt{\frac{1}{\gamma_{s1} - \gamma_{s2}}}$ ,  $\frac{\gamma_{s2}}{\gamma_{s1} - \gamma_{s2}}$ 

\nNow  $\gamma = \sqrt{\frac{1}{\gamma_{s1} - \gamma_{s2}}}$ ,  $\frac{\gamma_{s2}}{\gamma_{s2} - \gamma_{s1}}$ 

\nThus,  $\sqrt{\frac{1}{\gamma_{s1} - \gamma_{s2}}}$ ,  $\frac{\gamma_{s2}}{\gamma_{s1} - \gamma_{s2}}$ 

\nThus,  $\sqrt{\frac{1}{\gamma_{s1} - \gamma_{s2}}}$ ,  $\frac{\gamma_{s2}}{\gamma_{s2} - \gamma_{s1}}$ 

\nThus,  $\frac{\gamma_{s1} - \gamma_{s2}}{\gamma_{s2} - \gamma_{s2}}$ 

\nThus,  $\frac{\gamma_{s1} - \gamma_{s2}}{\gamma_{s1} - \gamma_{s2}}$ 

\nThus,  $\frac{\gamma_{s2} - \gamma_{s1}}{\gamma_{s2} - \gamma_{s1}}$ 

\nThus,  $\frac{\gamma_{s1} - \gamma_{s2}}{\gamma_{s2} - \gamma_{s2}}$ 

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